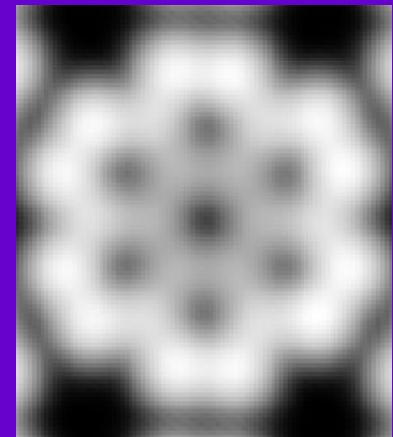


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# STM simulations as a strucutral tool

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**Jorge Iribas Cerdá**  
ICMM-CSIC (Spain)



## Outline

- STM theoretical concepts
- Applications

# Surface Studies by Scanning Tunneling Microscopy

G. Binning, H. Rohrer, Ch. Gerber, and E. Weibel

IBM Zurich Research Laboratory, 8803 Rüschlikon-ZH, Switzerland

(Received 30 April 1982)

Surface microscopy using vacuum tunneling is demonstrated for the first time. Topographic pictures of surfaces on an *atomic scale* have been obtained. Examples of resolved monoatomic steps and surface reconstructions are shown for (110) surfaces of  $\text{CaIrSn}_4$  and Au.

PACS numbers: 68.20.+t, 73.40.Gk

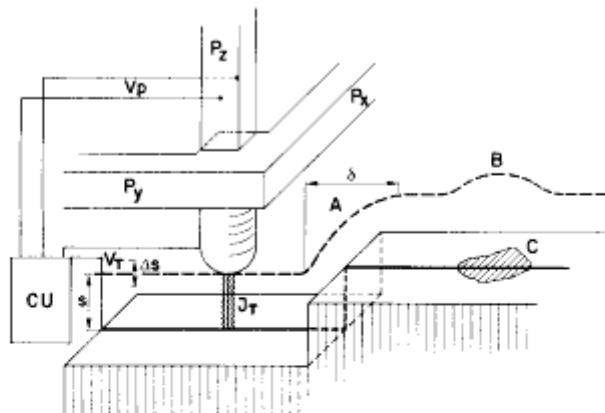


FIG. 1. Principle of operation of the scanning tunneling microscope. (Schematic: distances and sizes are not to scale.) The piezodrives  $P_x$  and  $P_y$  scan the metal tip  $M$  over the surface. The control unit (CU) applies the appropriate voltage  $V_p$  to the piezodrive  $P_z$  for constant tunnel current  $J_T$  at constant tunnel voltage  $V_T$ . For constant work function, the voltages applied to the piezodrives  $P_x$ ,  $P_y$ , and  $P_z$  yield the topography of the surface directly, whereas modulation of the tunnel distance  $s$  by  $\Delta s$  gives a measure of the work function as explained in the text. The broken line indicates the  $z$  displacement in a  $y$  scan at (A) a surface step and (B) a contamination spot,  $C$ , with lower work function.

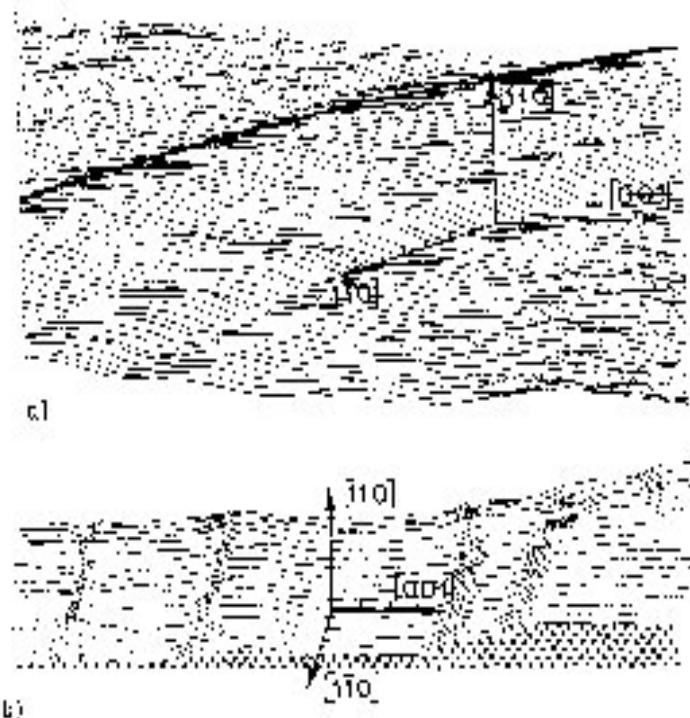


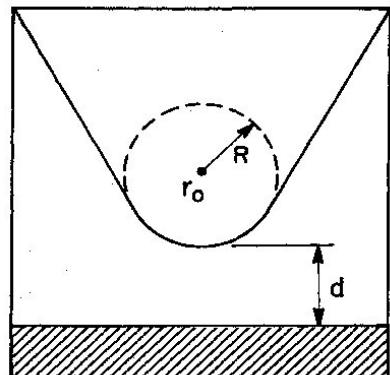
FIG. 3. Two examples of scanning tunneling micrographs of a Au (110) surface, taken at (a) room temperature, and (b) 300°C after annealing for 20 h at the

## Theory of the scanning tunneling microscope

J. Tersoff\* and D. R. Hamann

*AT&T Bell Laboratories, Murray Hill, New Jersey 07974*

(Received 25 June 1984)



- Elastic Scattering
- Weak tip-sample interaction:
  - Potential drop localized at vacuum region
  - First-order approximation for current  
(Bardeem approximation)
- s-wave model for the tip

**STM does NOT image atoms,  
but maps charge density at the tip position &  $E_f$**

### GOODS

- Works fine for many systems
- Allows DFT formalisms
- Reproduces bias dependence in semiconductors

### BADS

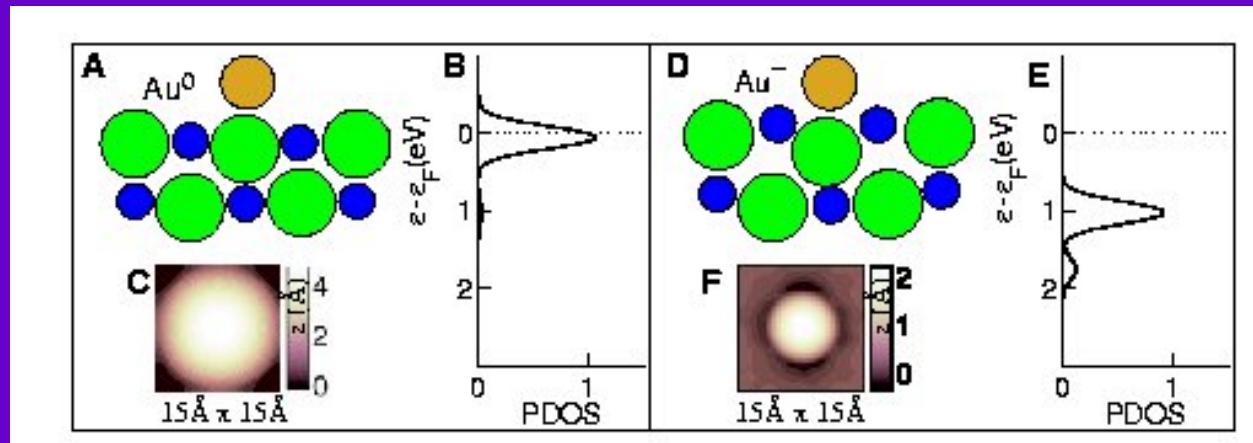
- No tip features
- Not fully reliable
- Wrong predictions ...  
(atomic resolution?)

# Tersoff-Hamann theory Application to single Ag

---

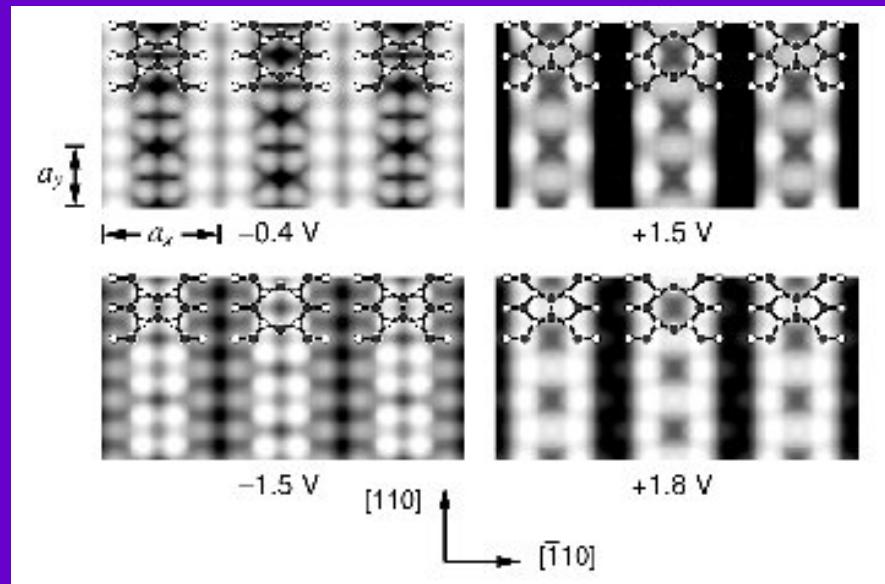
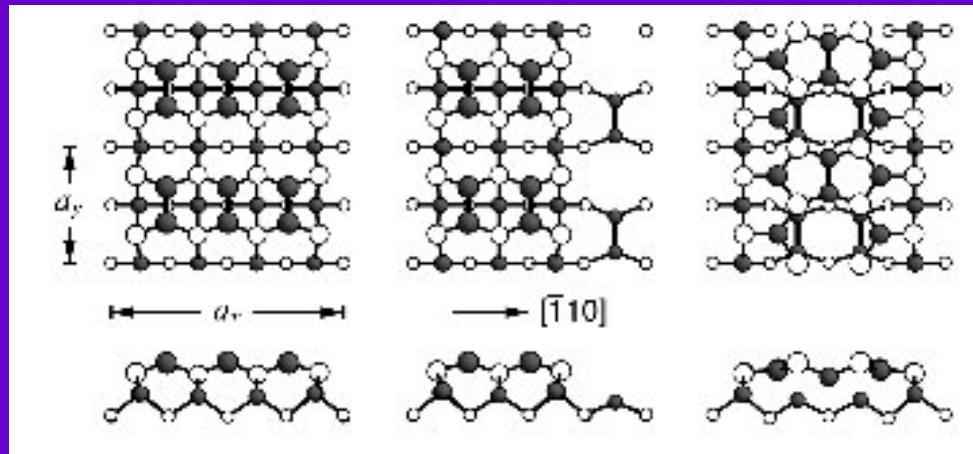


J. Repp *et al*, *Science* (04)



# Tersoff-Hamann theory Application to GaAs(100)-c(8x2)

S.-H. Lee et al,  
PRL (00)



# STM Simulations: *green*

[www.icmm.csic.es/jcerda](http://www.icmm.csic.es/jcerda)

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- Geometry
- Elastic Scattering
- k-sampling
- Tunneling regime
- EHT approximation

# Structural Sensitivity

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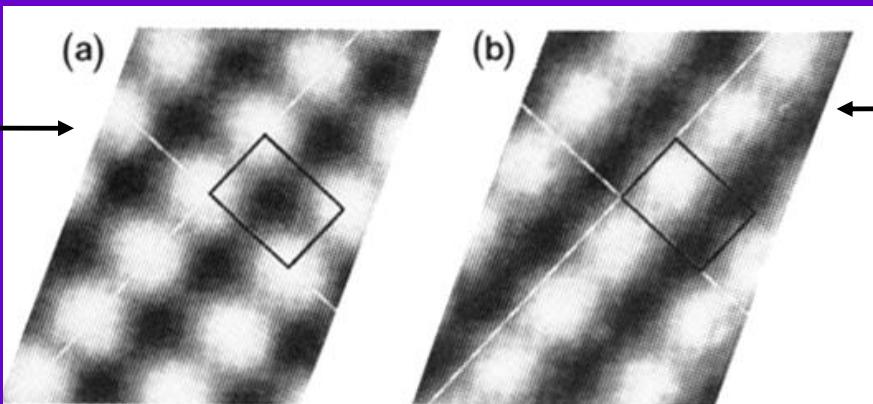
**Very Variable !!!**

- Experiment vs. Theory comparison:
  - Image aspect
  - Corrugations
- Theoretical variables to play with:
  - Surface structure (DFT)
  - Tip: nature, structure, orientation, etc.

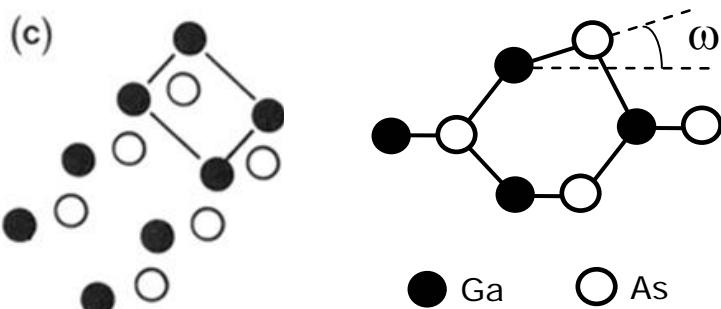
# GaAs(110) Surface

R.M. Feensrta et al, PRL (87)

Empty states  
Ga atoms

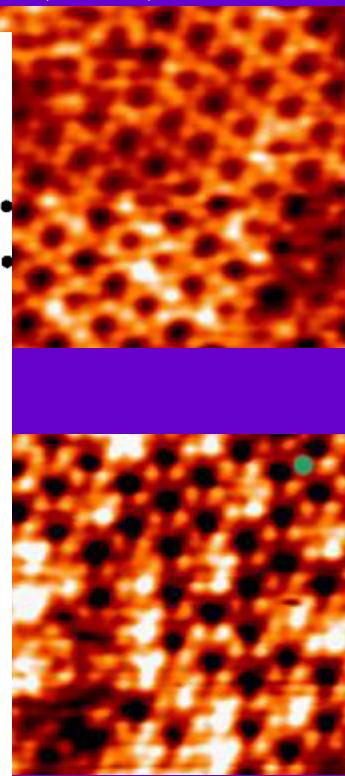
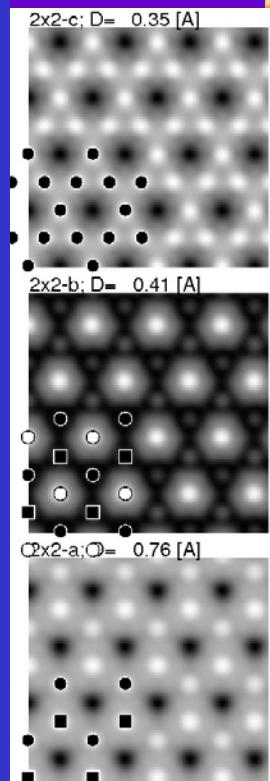


Filled states  
As atoms



# Rh(111)+nNO (Tip matters)

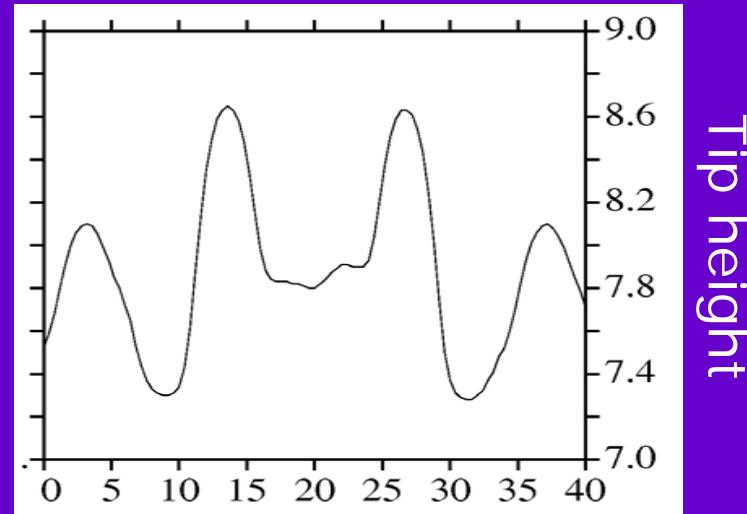
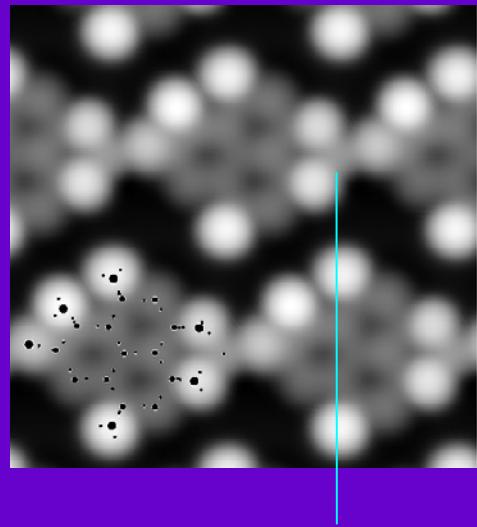
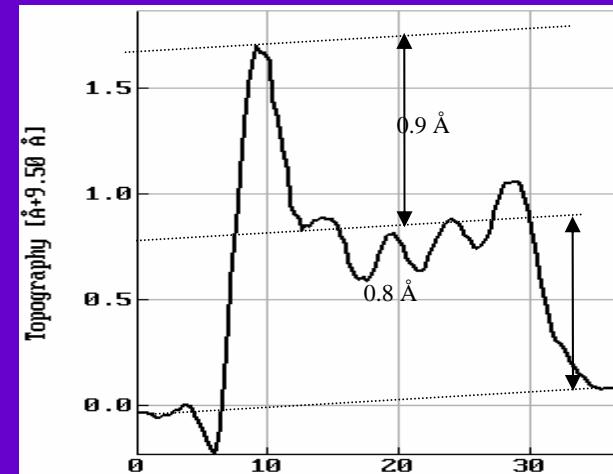
W(111)+NO (2x2)+3NO



(2x2)+2NO

# Pd(111)+H<sub>2</sub>O: Low T Water clusters (Corrugation matters)

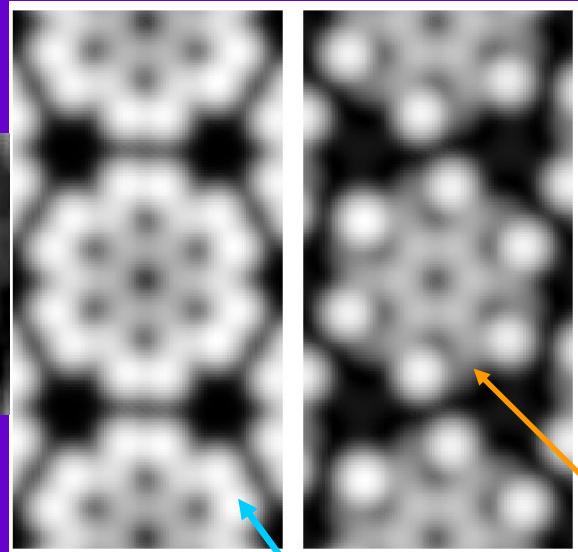
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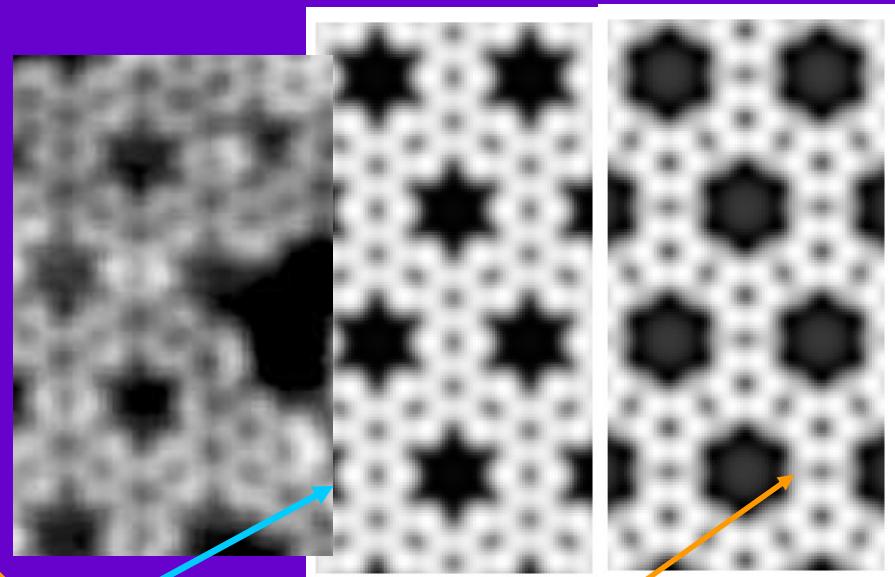
# Pd(111) + H<sub>2</sub>O (Water 2D quasiperiodic structures)

Rosettes

Lace



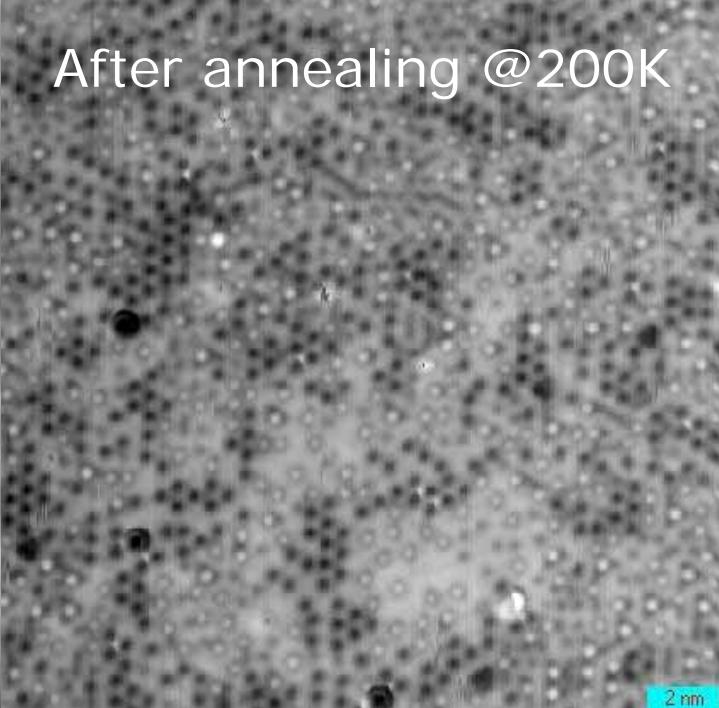
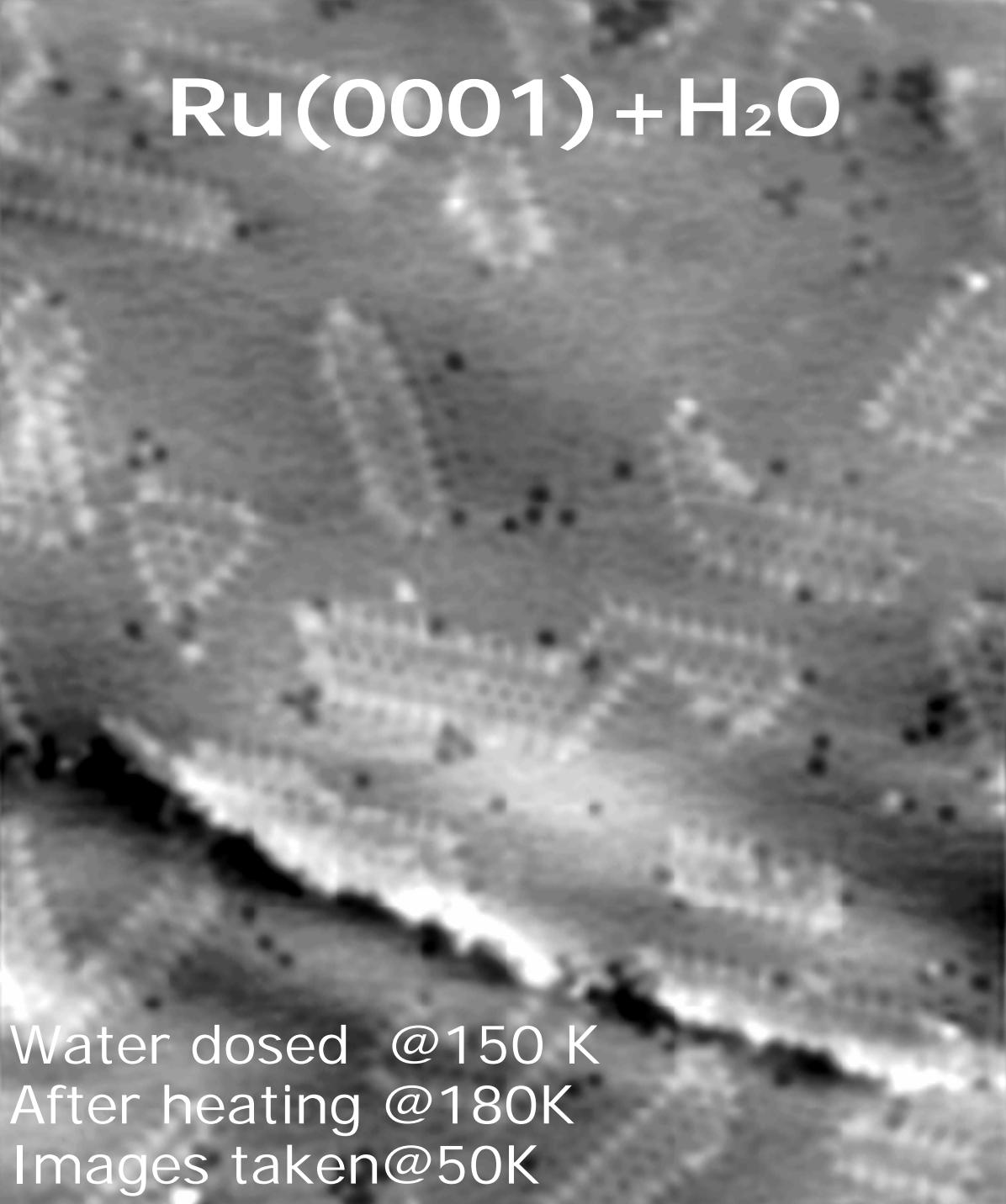
H<sub>2</sub>O 3D



OH groups

Ru(0001) + H<sub>2</sub>O

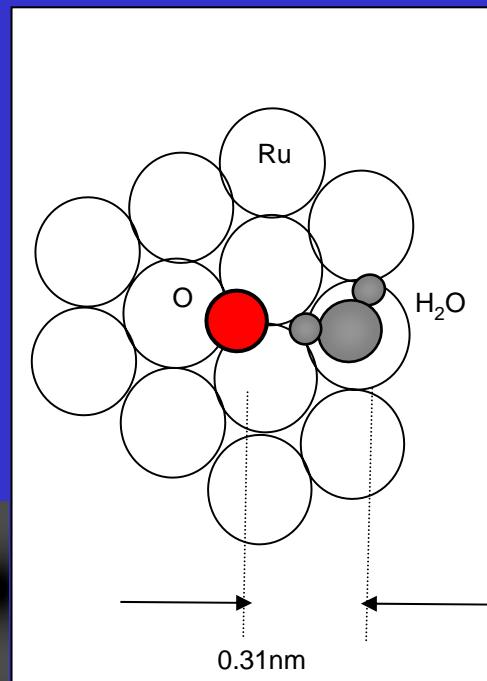
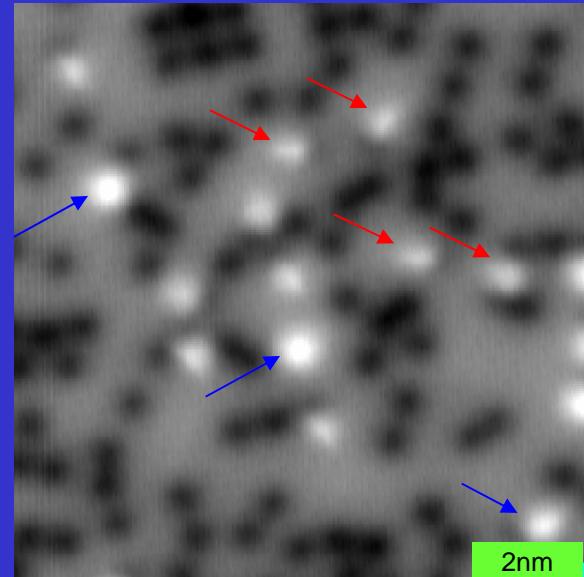
After annealing @200K



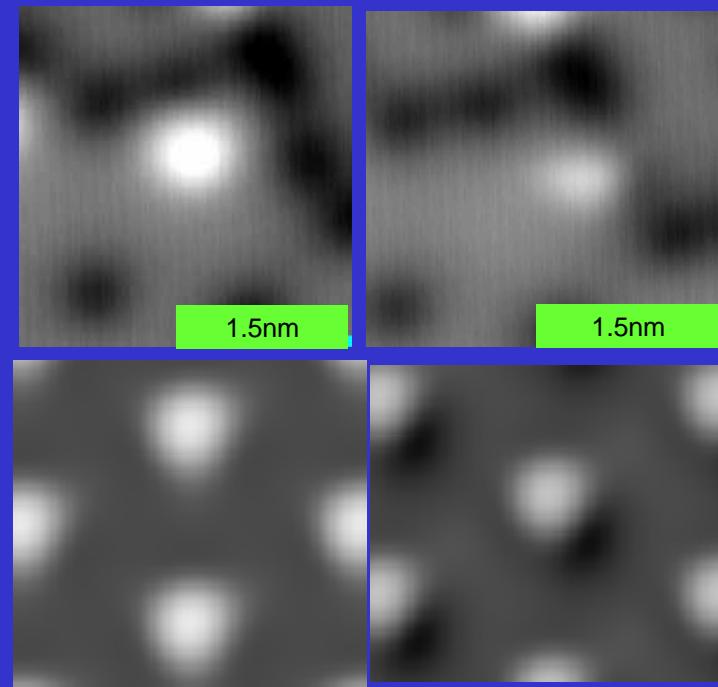
Water dosed @150 K  
After heating @180K  
Images taken@50K

1 nm

# Species before Water Desorption (surface oxygen precovered)



Experiment

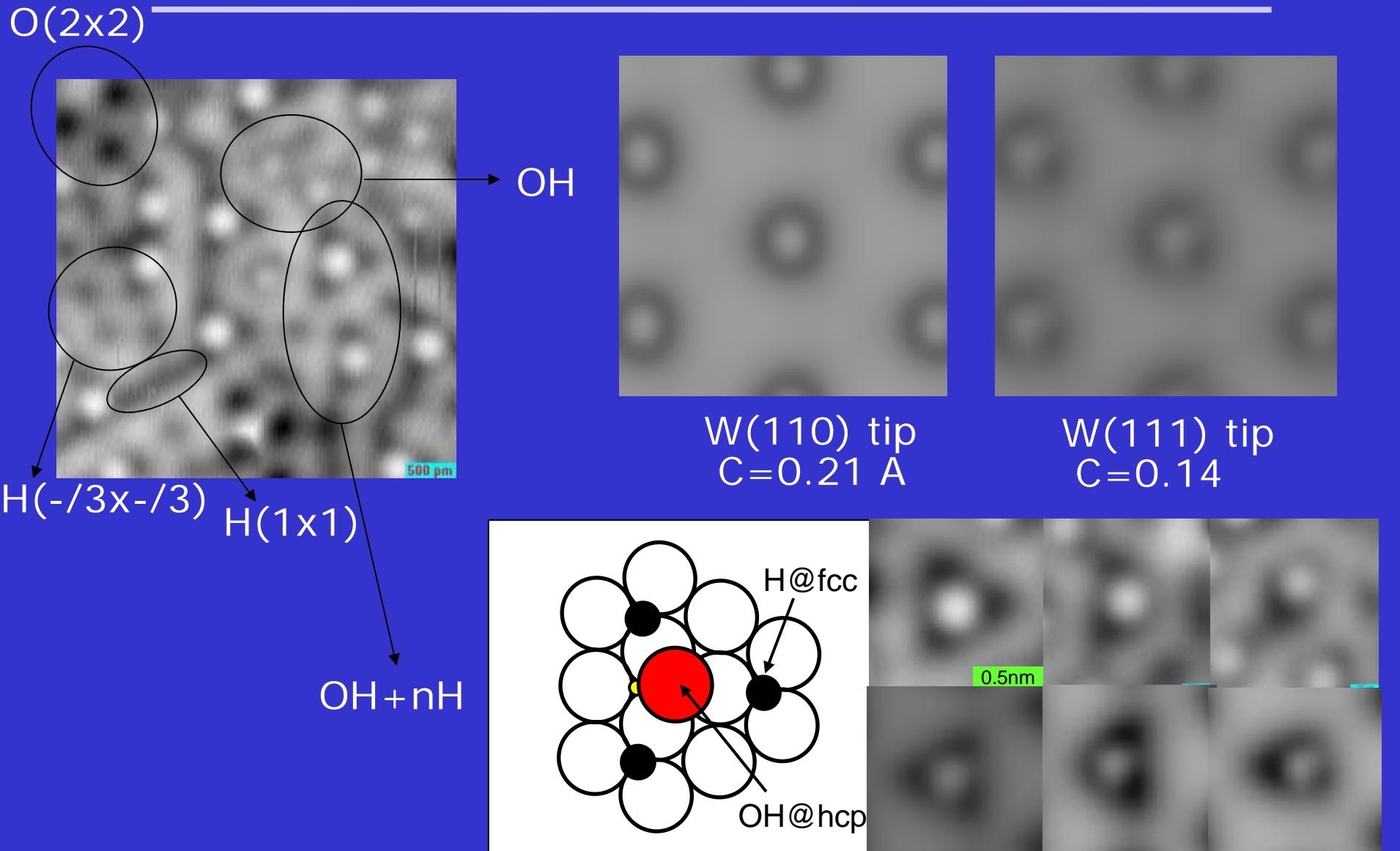


Theory

W(110) tip  
C=0.43 Å

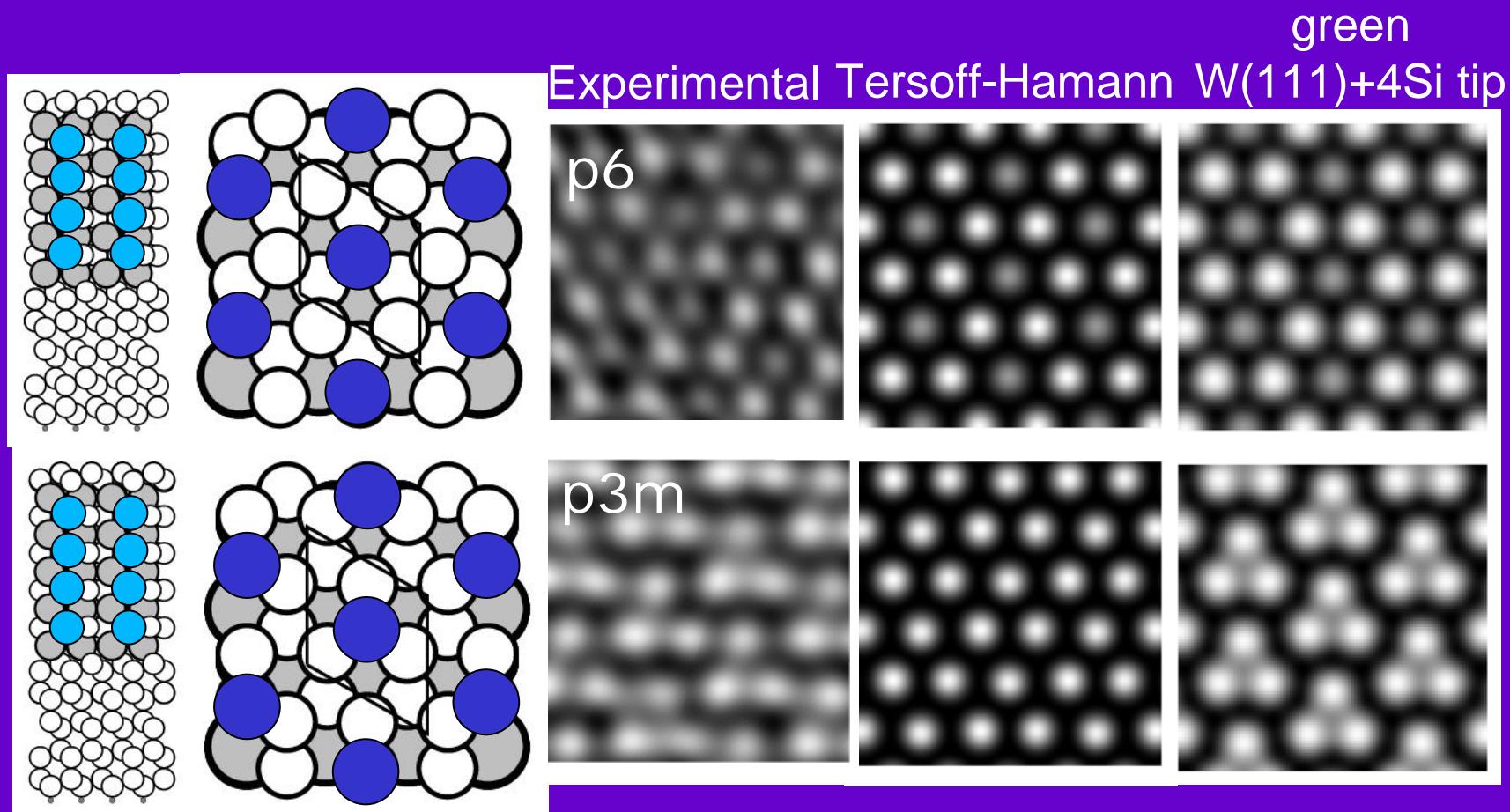
W(111) tip  
C=0.29

# Species after Water Desorption (surface oxygen precovered)



# $\text{Si}(111)-(-/3x-/3)\text{R}30 + \text{Y}_3\text{Si}_5 + 2\text{Si}$

## STM depth sensitivity

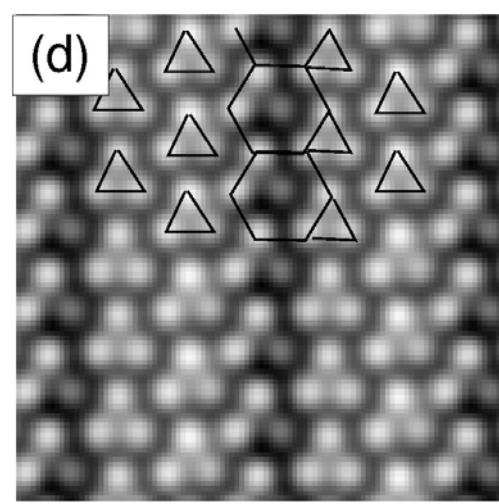
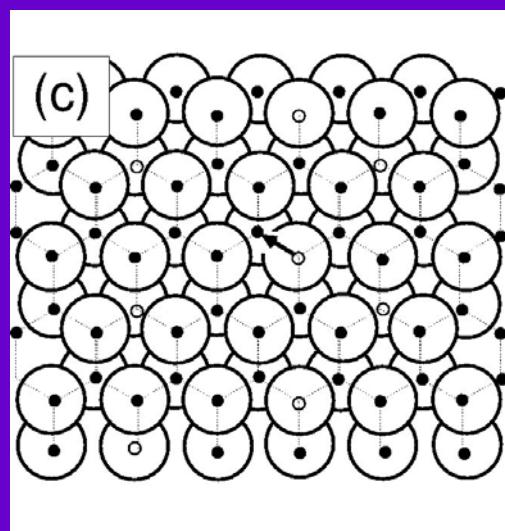
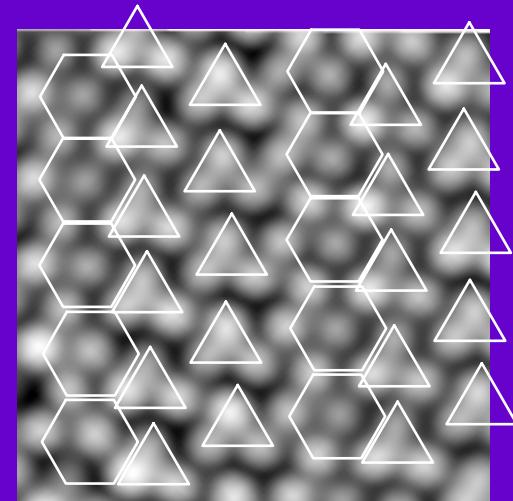
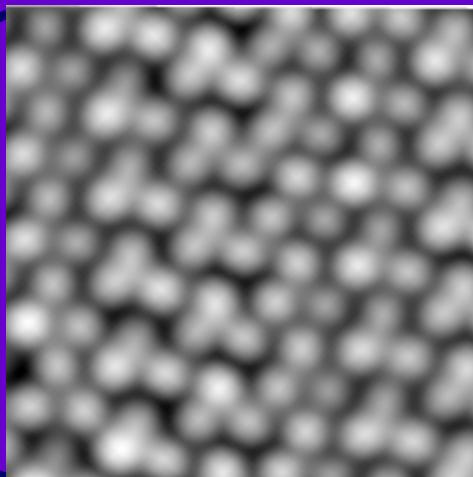
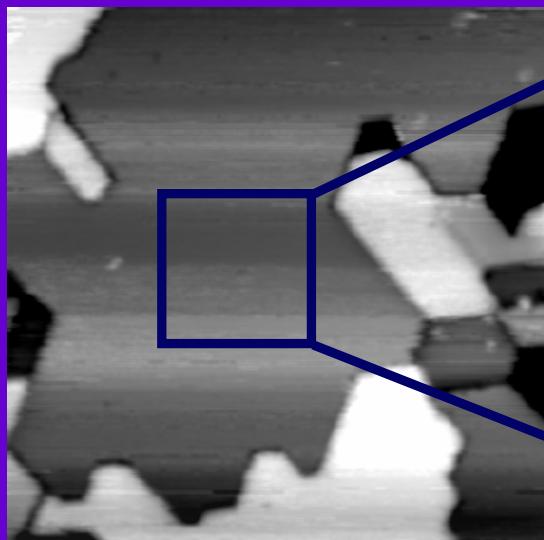


Vacancies at 3<sup>rd</sup> layer (almost 5 Å deep) dictate image aspect despite system is metallic!!

# Si(111)-(-/3x-/3)R30+Y<sub>3</sub>Si<sub>5</sub>

## Imaging Buried domain walls

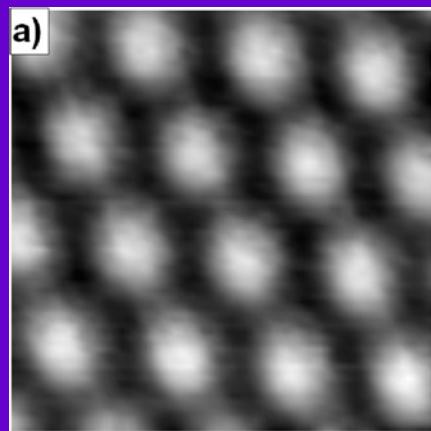
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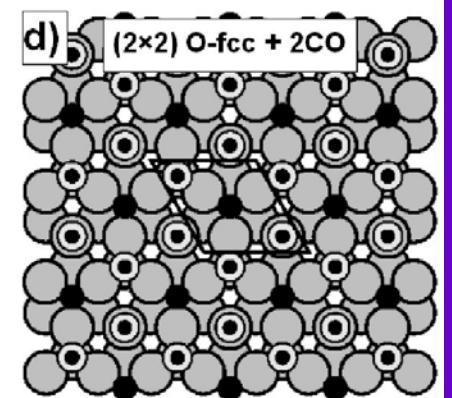
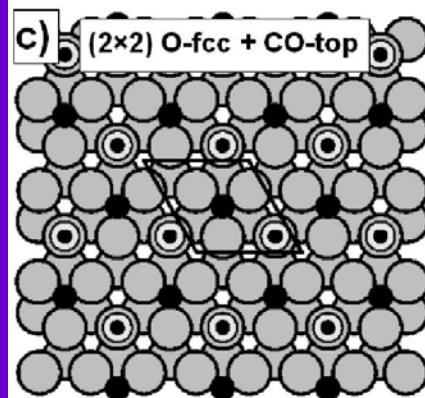
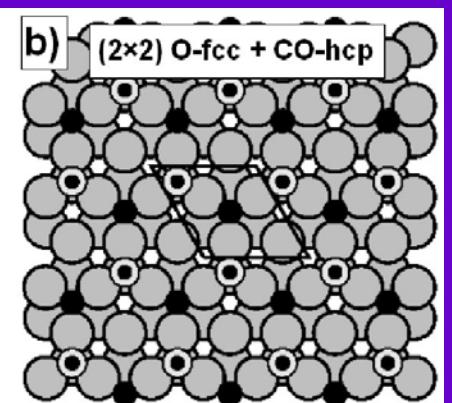
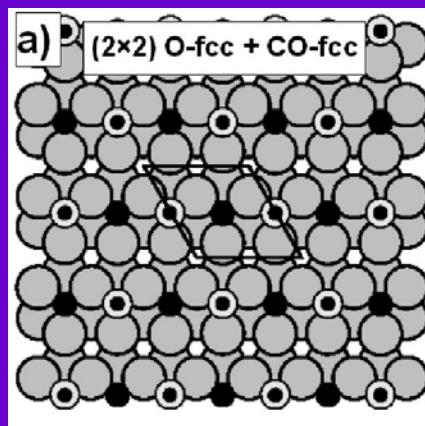
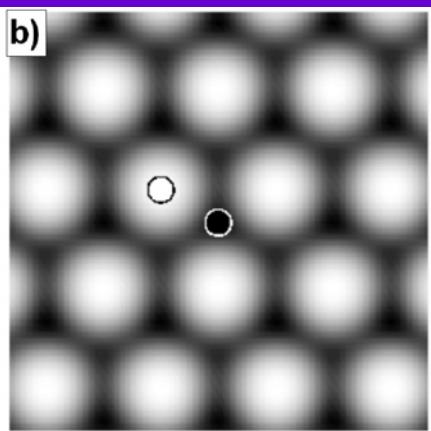
# Pd(111)-p(2x2)+CO+O

---

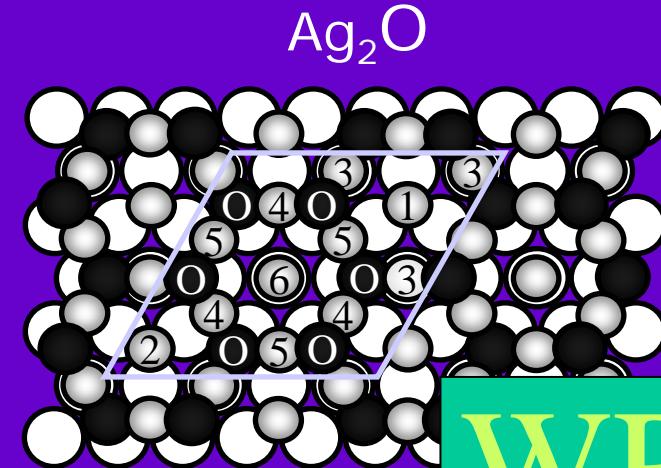
Experiment



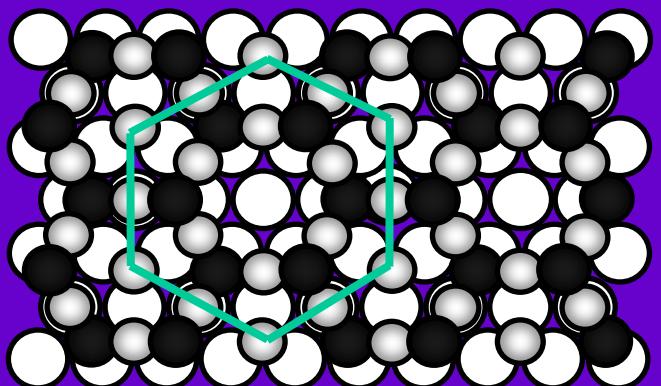
Theory



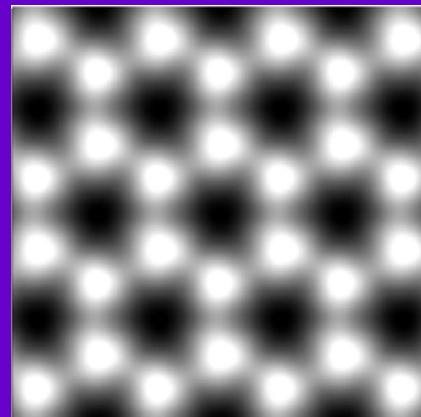
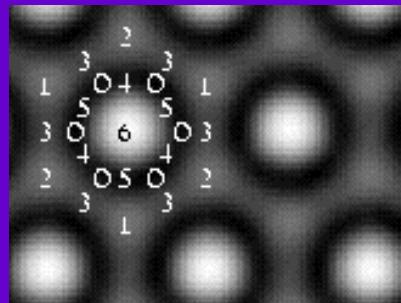
# Ag(111)+p(4x4)Ag<sub>2</sub>O



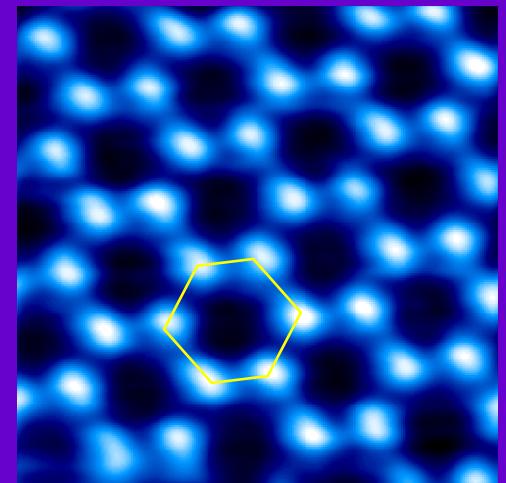
WRONG?



Simulations



Experimental



C.I. Carlisle et al,  
PRL (00)

# Conclusions

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- STM simulations is a “mature” field:
  - Most of the experimental features can be well reproduced
- STM simulations can be a very useful tool for structure analysis, but not always !!
  - Different models may yield the same image
- Take care with:
  - Convergence parameters: # atoms/k-points
  - Tip modelling
  - Using normal exp. images for comparison

# Acknowledgements

---

Theory: **M.-L. Bocquet , P. Sautet**

IRC-CNRS, Villerbaune & ENS de Lyon

Pd+H<sub>2</sub>O: **A.Michaelides** (Cambridge University, UK)

**P. J. Feibelman** (SNL, Albuquerque, USA)

**T. Mitsui, E. Fomin, M. Rose and M. Salmeron**

(LBNL, Berkeley, USA)

Ru+H<sub>2</sub>O: **A. Mugarza, T. Shimizu, D. Ogletree, M. Salmeron**

(LBNL, Berkeley, USA)

Ag<sub>2</sub>O: **C.I. Carlisle, M. Webb, A.Alavi & D. King**

(Cambridge University, UK)

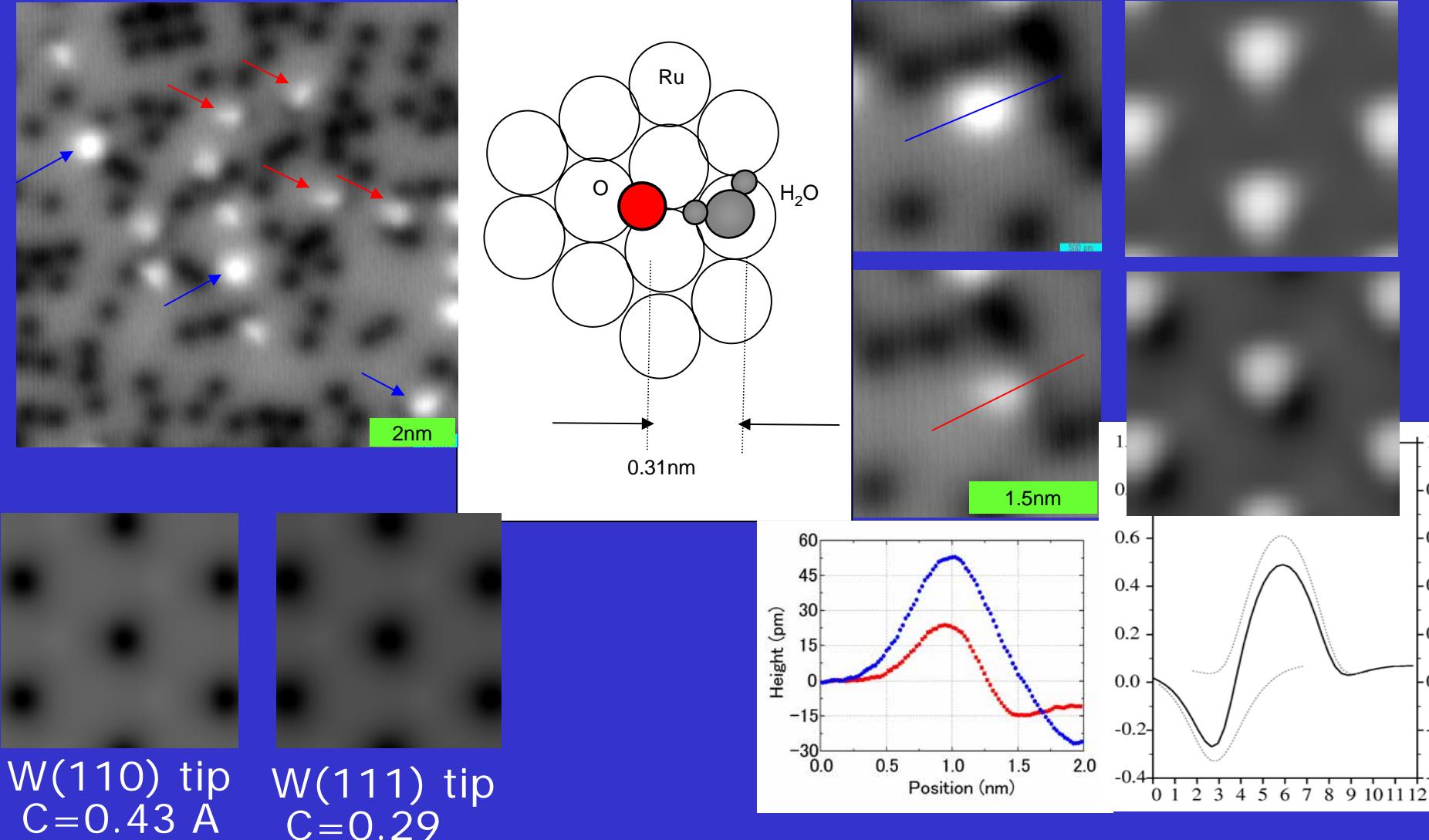
CO+O: **J. Méndez** (ICMM-CSIC)

**S.H. Kim, J. Wintterlin, G. Ertl** (FHI, Berlin)

Y<sub>3</sub>Si<sub>5</sub>: **C. Rogero, J.A. Martín Gago** (ICMM-CSIC)

Rh+NO: **K. Flipse, C. Poppe and J. Haagelard** (Holland)

# Species before Water Desorption (surface oxygen precovered)



# Green's Functions

---

**Definition:**

$$F(E, k_{\text{par}}) = E^* S(k_{\text{par}}) - H(k_{\text{par}})$$

$$G(E, k_{\text{par}})^* F(E, k_{\text{par}}) = I$$

**Projection onto PLs i,j:**

$$G_{ij} = (F^{-1})_{ij}$$

$$T_{ij} = G_{ij} (G_{jj})^{-1}$$

**Sequential Stacking (Dyson):**

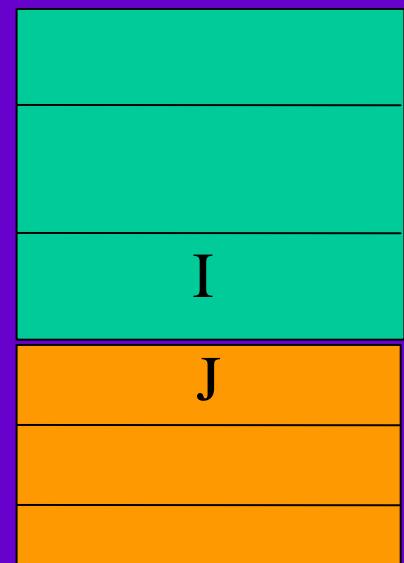
$$T_{IJ} = -G_{II}^0 F_{IJ}$$

$$\Sigma_{JJ} = F_{JI} T_{IJ}$$

$$G_{JJ} = (G_{JJ}^0 + \Sigma_{JJ})^{-1}$$

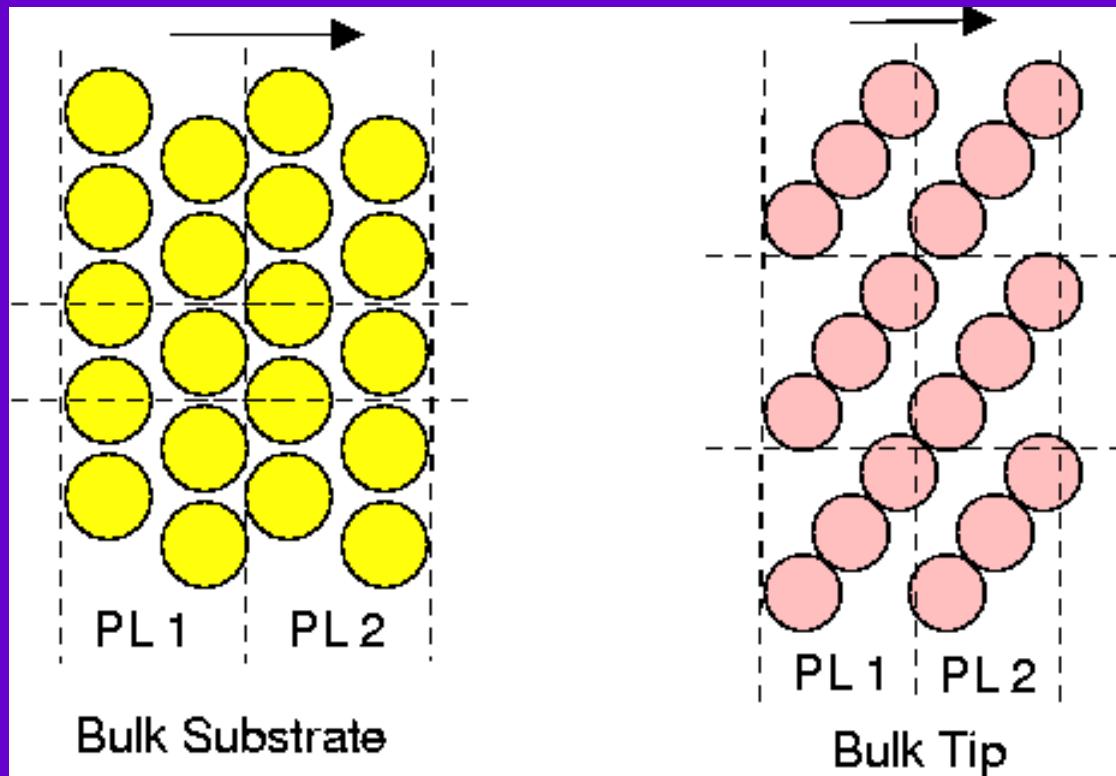
**SGFM (Molinier):**

$$G_{ij} = G_{ij}^0 + T_{ii}^0 (G_{IJ} - G_{IJ}^0) T_{jj}^0$$



# Bulk Stacking

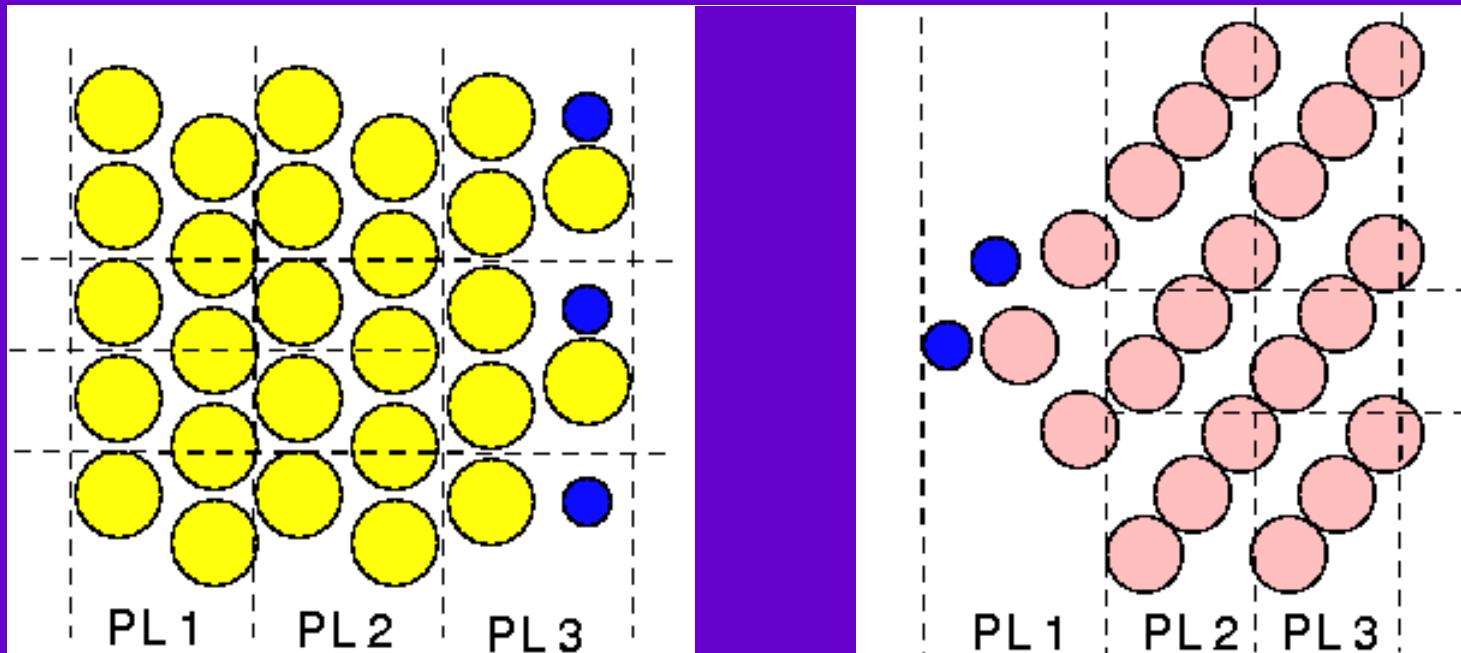
---



$-G_{b+/-}$  ,  $T_{b+/-}$  ,  $k_{\text{perp}}(E)$  ,  $U_{k\text{perp}}$

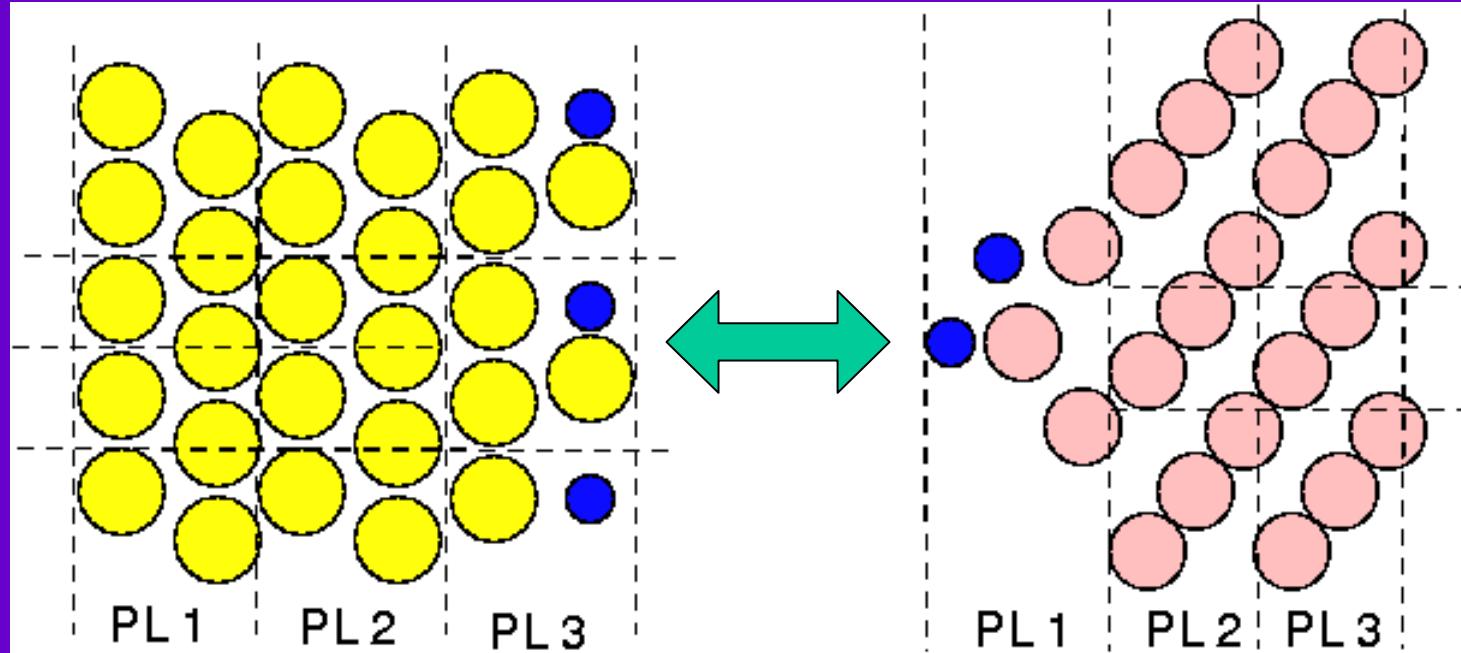
# Surfaces

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Surface (Localized) States?

# Tip-Sample Interaction



First Order:  $G_{JJ} = G^0_{JJ} \Rightarrow G_{IJ} = T_{IJ} * G^0_{JJ}$

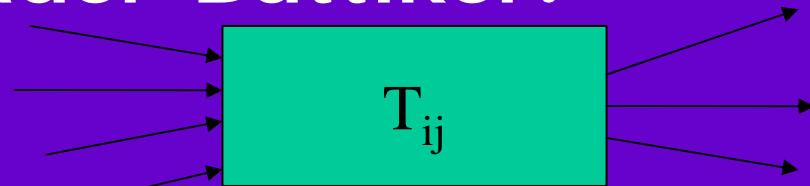
2nd order tunneling events neglected

# Current Formula

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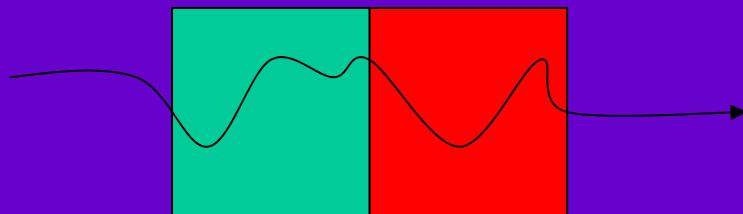
$$I(V) = \sum_{k\text{par}} \int dE \sigma(E, V, k\text{par}) f_{\text{tip}}(E - eV) f_{\text{subs}}(E)$$

-Landauer-Büttiker:



Links eigenstates at the bulk electrodes

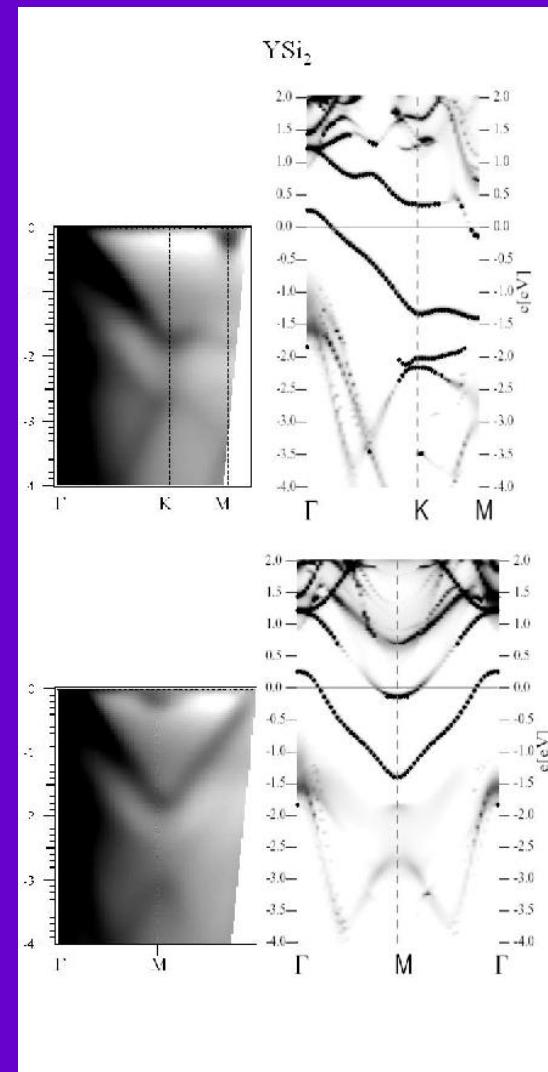
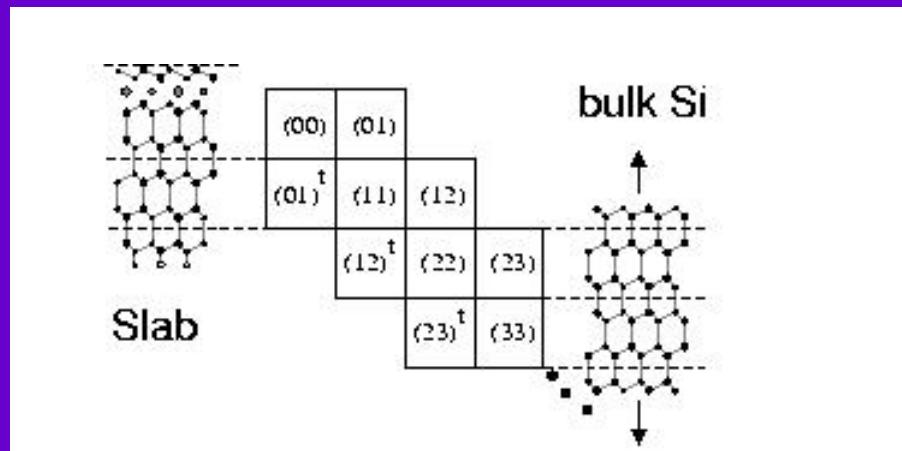
Green's Functions (Todorov,Datta):



Subspace of propagative states

# DFT (siesta)

- Do self-consistency by pieces



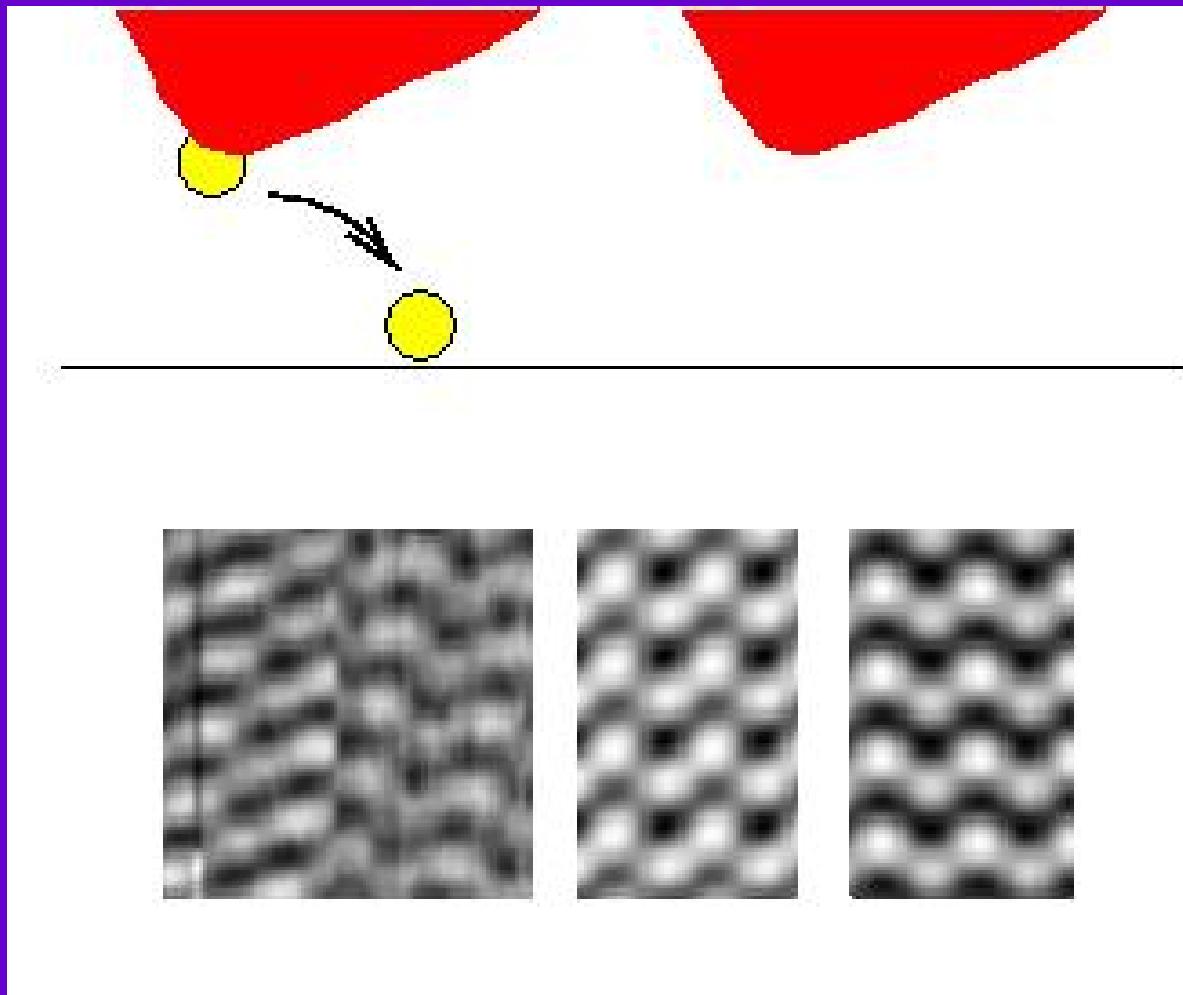
# Some Key Issues

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- k-sampling
- $E_i$ 
  - Required for integration but induces damping
  - Want to skip surface states?
- $\mathbf{I}$  :  $N_E \times N_{PL} \times N_{kc} \times (N_{AO}, N_G)^2$ 
  - # atoms > 1000 ;  $N_{AO} \times N_G < 1000 \Rightarrow$  Fast
- $\mathbf{DM}$  :  $N_E \times N_{PL}^2 \times N_{kc} \times (N_{AO}, N_G)^2$
- Loops: bias , energy , spin , pixels,  $z_{tip}$  , topography
  - $\Rightarrow$  save/allocate dynamically
- AO basis small  $\Rightarrow$  Corrugations within a factor 2

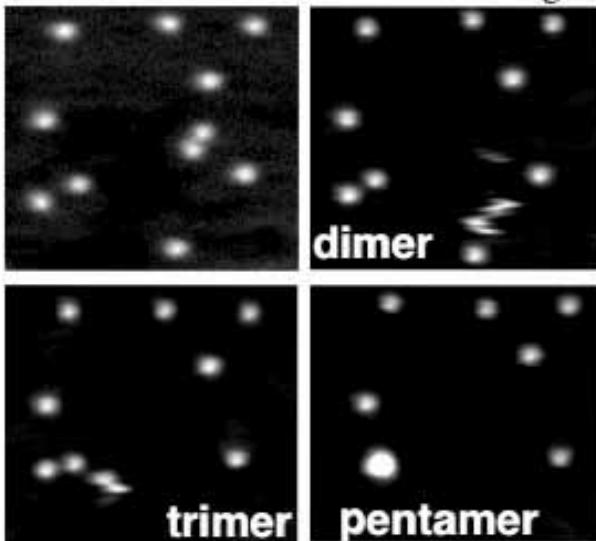
# Tip effects

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# Water on Pd(111)

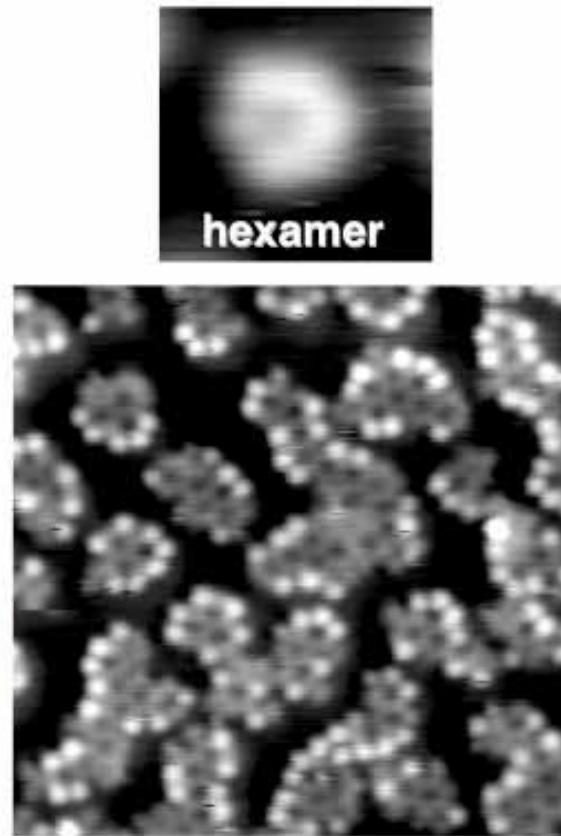
Monomer diffusion and clustering



hexamer

## diffusion coefficient

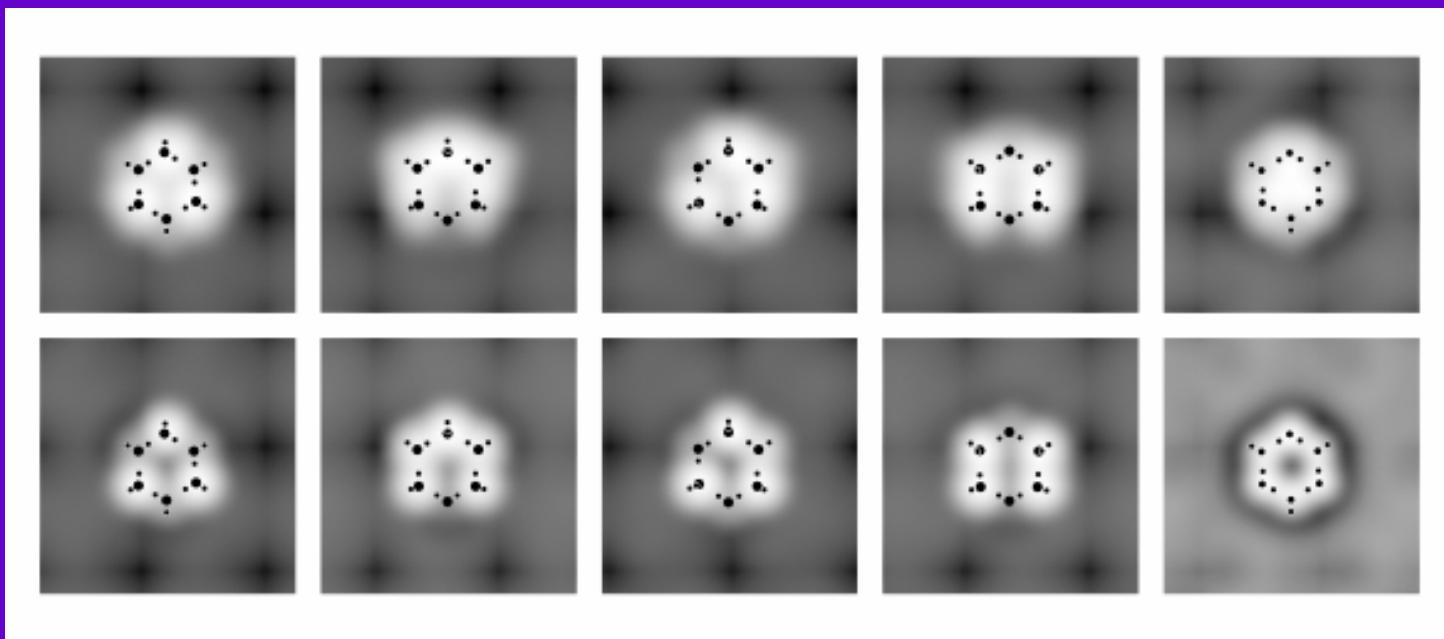
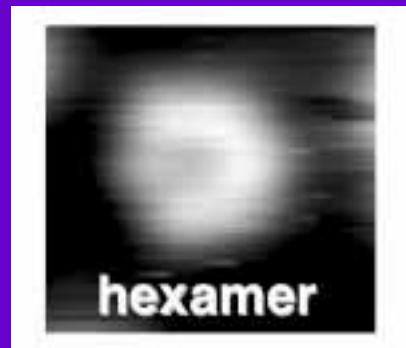
- monomer ~  $2.30 \times 10^{-3} \text{ Å}^2/\text{s}$
- dimer >  $50 \text{ Å}^2/\text{s}$
- trimer, tetramer ~  $1.02 \text{ Å}^2/\text{s}$



Hexamers forming honeycombs at 40K.  
Bright molecules at edges:  
dangling H bonds ?

# Hexamers

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W(111)  
tip

Pt(111)  
tip